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Characterization of Cr³⁺ centres in LiNbO₃ using fluorescence line narrowing

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Abstract. The optical spectroscopic properties of Cr³⁺ in LiNbO₃ have been investigated using optical absorption, fluorescence, fluorescence line narrowing (FLN), selective excitation and radiative lifetime measurements. The present results are compared with data obtained using other techniques including electron spin resonance (ESR) and electron nuclear double resonance (ENDOR), and shown to be consistent with there being five Cr³⁺ centres. The predominant centre is due to substitution at the Li⁺ site, which provides the weak crystal field in which Cr³⁺ ions emit into a broad ${}^4T_2 \rightarrow {}^4A_2$ band. Using FLN, two centres experiencing a stronger crystal field are shown to be due to substitution at this same site, probably perturbed by the location of Nb⁵⁺ antisites and Nb⁵⁺ vacancies in the next-nearest-neighbour cation shell.

The addition of Mg²⁺ ions to LiNbO₃ appears not to create new Cr³⁺ complexes: rather it modifies the nature of the disorder, thereby changing the relative concentrations of the different centres. To this extent it becomes possible to identify two other lightly occupied Cr³⁺ centres where substitution is at the Nb⁵⁺ site. This total of five Cr³⁺ centres in LiNbO₃ gives a consistent interpretation of all the available spectroscopic evidence, including ESR and ENDOR, while pointing to the difficulties of theoretical work in sites where the differences in the energy levels of Cr³⁺ ions are too small to be distinguished by the available models.

1. Introduction

Cr³⁺ ions in lithium niobate (LiNbO₃) emit broad band luminescence when excited with visible light, and this observation has stimulated many recent spectroscopic studies of Cr³⁺:LiNbO₃. These investigations have revealed that the material also fluoresces via a number of much narrower R lines. The results of such studies are not fully understood and different researchers have drawn conflicting conclusions.

It is well known that in strong-field sites at low temperature the luminescence of Cr³⁺ ions is predominantly via the ${}^2E \rightarrow {}^4A_2$ transition (R lines), whereas in weak-field sites the broad-band ${}^4T_1 \rightarrow {}^4A_2$ transition is dominant. It should be noted that the difference in crystal-field strength between weak- and strong-field sites may actually be very small; the distinction is between sets of ions experiencing an intermediate crystal field close to the crossing between 4T_2 and 2E states. For the purposes of this paper, a strong crystal field is defined as resulting in the 2E state being lower than the 4T_2 state and vice versa for weak crystal fields. The simultaneous presence of several R lines and broad-band fluorescence suggests that Cr³⁺ ions substitute into multiple environments in LiNbO₃.

The structure of LiNbO₃ below its ferroelectric Curie temperature belongs to the R3c space group [1]. Shown schematically in figure 1 are the three interlocking octahedral complexes that repeat to compose the crystal structure. These three complexes are each defined by the central ion (or void) and are stacked along the *c* axis in the sequence

Li, Nb, empty octahedron, Li, Nb, empty octahedron, etc, each separated by planes of oxygen ions. The symmetry at the Li^+ and Nb^{5+} sites and the empty octahedron is C_3 , although C_{3v} is often regarded as a good approximation [2, 3]. All three sites are candidates for occupancy by the dopant Cr^{3+} ion given the preference this ion has for octahedral coordination. Although stoichiometric crystals are grown, it is easier to grow LiNbO_3 crystals with the congruent composition, $(\text{Li}_{1-5x}\text{Nb}_{5x})\text{Nb}_{1-4x}\text{O}_3$, where $x \approx 0.0118$. This formula implies that almost 6% of Li^+ sites are occupied by Nb^{5+} ions (referred to as Nb antisites) and that an appropriate number of Nb^{5+} sites are vacant to ensure charge neutrality [1, 4].

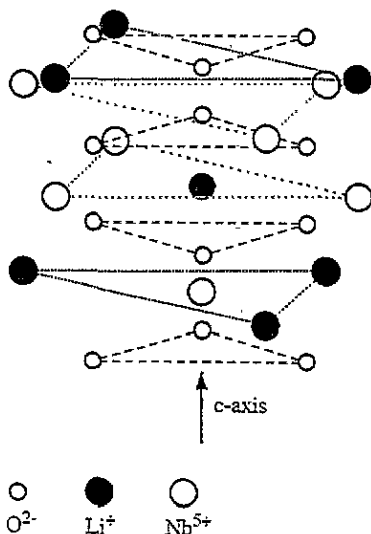


Figure 1. A schematic representation of the structure of LiNbO_3 . The central Li^+ site is surrounded by an approximate octahedron of nearest-neighbour O^{2-} ions, an approximate octahedron of second-nearest-neighbour Nb^{5+} ions and a slightly larger approximate octahedron of Li^+ ions. Interlocking octahedra of oxygen ions are defined by the occupation of their central position and follow along the c axis in the order empty octahedron, Li^+ , Nb^{5+} . The guide lines on the figure indicate triangles of like ions in the ab plane.

In order to make LiNbO_3 more resistant to damage in frequency doubling applications it is common practice to add MgO to the melt, thus causing an additional source of disorder within the lattice. The non-stoichiometry of the crystals, the influence of Mg^{2+} codoping and the many possibilities for charge compensation that result from Cr^{3+} ions substituting into a lattice with no triply ionized host ions lead to complicated optical and magnetic spin resonance spectra for $\text{Cr}^{3+}:\text{LiNbO}_3$ and $\text{Cr}^{3+}:\text{Mg}^{2+}:\text{LiNbO}_3$. First reports of these spectra appeared in 1966 [5] and have remained the subject of much debate.

The objective of the present study is to resolve the contradictory conclusions that have been drawn by various workers. High-resolution laser spectroscopic measurements of $\text{Cr}^{3+}:\text{LiNbO}_3$ and $\text{Cr}^{3+}:\text{Mg}^{2+}:\text{LiNbO}_3$ have been conducted to make possible a definitive interpretation of the optical spectra. These results have also been used to interpret the observed optical spectra in a manner consistent with all the published ESR and ENDOR results. This has made it possible to develop a realistic model of the site occupancy of Cr^{3+} ions in LiNbO_3 and MgO-LiNbO_3 .

2. Experimental methods

Optical absorption spectra were measured using an AVIV spectrophotometer that operated in the wavelength range 190–3000 nm. The samples were cooled to 77 K in a liquid nitrogen cryostat. In photoluminescence experiments the samples were cooled to 15 K in a cryorefrigerator. The broad-band luminescence was excited by a dye laser, the emitted radiation being dispersed in a 0.5 m monochromator and detected by a germanium detector. Corrections were made for the spectral response of the system.

In FLN studies the R lines were excited by the highly monochromatic output from a tuneable Ti:sapphire laser, which has a jitter-limited linewidth of less than 1 MHz. The luminescence was dispersed in a 1 m monochromator and detected using photon counting electronics. In order to measure the splitting of the ${}^4\text{A}_2$ ground state, higher-resolution spectra were obtained by analysing the radiation emitted by the sample with a scanning Fabry–Perot etalon, placed before the 1 m grating monochromator to isolate the specific spectral range of interest. The etalon plates were set 1 mm apart giving a free spectral range of 5 cm^{-1} with a transmission linewidth of 0.1 cm^{-1} . In these FLN studies the wavelength of excitation was measured using a wavemeter. The R line luminescence in LiNbO_3 is very weak; the short lifetime and the necessity to use a monochromator for auxiliary dispersion meant that time-resolved spectra composed of resonant and non-resonant lines could not be obtained. The ground state splittings of the R lines were therefore determined using cw excitation, the strong laser scatter in the resulting spectrum marking the resonant condition whilst much weaker non-resonant features correspond to fluorescence from the sample decaying to the alternative ground state level.

3. Results and discussion

3.1. General spectroscopic measurements

The optical spectroscopy of Cr^{3+} ions in LiNbO_3 was reported first by Burns *et al* [5]. They identified two sharp R lines in absorption at wavelengths of 724.2 nm and 726.2 nm due to ${}^4\text{A}_2 \rightarrow {}^2\text{E}$ transitions split by a trigonal field. They also observed the broad ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$, ${}^4\text{T}_1$ absorption bands spanning the visible region. The $\text{Cr}^{3+}:\text{LiNbO}_3$ sample used in this study gave similar absorption spectra (figure 2). Also shown in figure 2, superposed on the broad absorption bands, are sharper features associated with spin-forbidden transitions from the ${}^4\text{A}_2$ ground state to the ${}^2\text{T}_1$ and ${}^2\text{T}_2$ levels.

To check the validity of the transition assignments, crystal-field theory is applied to calculate the energies of the states. Sugano and coworkers [6] derived the energy matrices for the term splittings of 3d electrons in octahedral fields and these have been applied successfully to interpret the observed optical spectra of transition metal ions in many host crystals. The matrices depend on the strength of the octahedral crystal field, Dq , and the Racah parameters B and C . The peak energy of the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$ band is $10 Dq$, after defining the ground state ${}^4\text{A}_2$ to be the energy zero. Diagonalizing the 2×2 matrix of the ${}^4\text{T}_1$ terms to fit the peak of the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_1$ band yields the B parameter, and the C parameter is obtained from the energy of the ${}^2\text{E}$ level. The energies of the other levels are then calculated by diagonalizing the appropriate matrices. The calculated and observed energies given in table 1 are in good agreement, although the results of these calculations are valid only for the dominant centre that is introduced below as the γ centre.

The broad luminescence band, excited in either the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$ band and peaking at 900 nm, was also first observed by Burns *et al* [5], and was tentatively identified with the

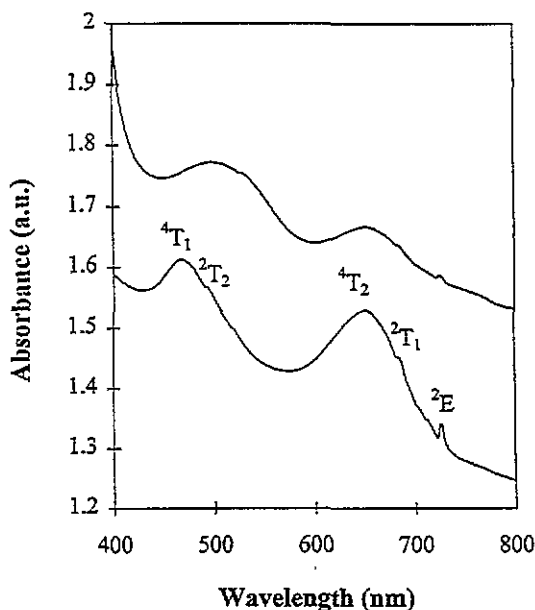


Figure 2. Optical absorption spectra of Cr^{3+} ions in LiNbO_3 (lower) and $\text{MgO}:\text{LiNbO}_3$ (upper) measured at room temperature.

Table 1. Calculated energies of multiplets of Cr^{3+} in LiNbO_3 using the octahedral approximation. $Dq = 1530 \text{ cm}^{-1}$, $C = 3204 \text{ cm}^{-1}$ and $B = 558 \text{ cm}^{-1}$.

Level	Calculated energy (cm^{-1})	Observed energy (cm^{-1})
^2E	13 798	13 790
$^2\text{T}_1$	14 254	14 340
$^4\text{T}_2$	15 300	15 300
$^2\text{T}_2$	20 640	20 200
$^4\text{T}_1$	21 000	21 000

Stokes-shifted $^4\text{T}_2 \rightarrow ^4\text{A}_2$ transition. In the present study it was found that the broad emission lineshape (figure 3(a)) is independent of excitation wavelength except at the low-energy wing of the $^4\text{A}_2 \rightarrow ^4\text{T}_2$ absorption band near 750 nm, where the resulting emission is red shifted by around 50 nm.

The fluorescence spectrum contains three sharp lines near 730 nm (figure 3(b)), two of which were first reported by Glass [7]. The integrated intensity of these lines is of the order of one-thousandth of that of the broad band. These lines have been identified with the $^2\text{E} \rightarrow ^4\text{A}_2$ transitions, resulting from two overlapping R line pairs due to ^2E states split by the combined effects of trigonal field and spin-orbit coupling [8, 9]. The line at 735.5 nm is an R_1 line whose R_2 partner at 730.4 nm is degenerate with an R_1 line having its partner at 727.0 nm. The wavelengths of these R lines are quite different from those measured in absorption. These results lead to the important conclusion that in LiNbO_3 there are three distinct Cr^{3+} environments, producing three distinct R line systems. For purposes of identification the lowest-energy doublet, at 735.5 nm and 730.4 nm, is designated the $\text{R}_{1,2}(\alpha)$ lines, the lines at 730.4 nm and 727.0 nm are the $\text{R}_{1,2}(\beta)$ lines and the R lines observed in absorption (726.2 nm and 724.2 nm) are termed $\text{R}_{1,2}(\gamma)$. The corresponding

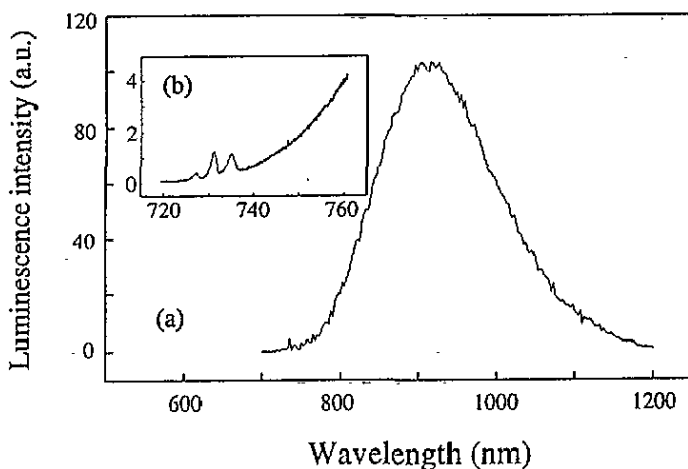


Figure 3. (a) The broad-band luminescence and (b) the R lines in emission from $\text{Cr}^{3+} : \text{LiNbO}_3$ measured at 15 K.

centres are referred to as α , β and γ , respectively. The γ centre emits from the ${}^2\text{E}$ level as $\text{R}_{1,2}(\gamma)$ lines at elevated temperatures through thermal depopulation of the ${}^4\text{T}_2$ level. Weiyi *et al* [8] attributed this emission to hot-phonon sidebands of the ${}^4\text{T}_2$ zero-phonon lines. However, in view of the similarity in energy, linewidths and splittings of these lines to the other R lines this explanation is unlikely. Evidence from excitation spectra completely rule out this interpretation.

Scanning the Ti:sapphire laser over the R line region in the absorption spectrum and detecting luminescence at 850 nm strongly picks out the $\text{R}_{1,2}(\gamma)$ lines, confirming the excitation spectrum published previously [8]. This demonstrates emphatically that the $\text{R}_{1,2}(\gamma)$ absorption lines and the 900 nm photoluminescence band originate from Cr^{3+} ions at the same weak-field site. Indeed, the entire optical spectrum of $\text{Cr}^{3+} : \text{LiNbO}_3$ is dominated by features associated with the γ centre, the R lines from centres α and β being almost undetectable in absorption and dwarfed by the broad band in luminescence (figure 3). In consequence it follows that the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$, ${}^4\text{T}_1$ absorption peaks in figure 2 also originate predominantly from the weak-field centre, γ .

The ${}^2\text{E} \leftrightarrow {}^4\text{A}_2$ transition is spin forbidden and becomes allowed through spin-orbit coupling to spin quartet states, mainly the ${}^4\text{T}_2$ state, which is energetically close. The transition moment is approximately

$$P_{eff} = \left({}^4\text{A}_2 \left| \mu_E \right| {}^4\text{T}_2 \right) \left({}^4\text{T}_2 \left| H_{so} \right| {}^2\text{E} \right) / \left[E({}^2\text{E}) - E({}^4\text{T}_2) \right] \quad (1)$$

where the dipole operator is μ_E and the effects of odd-parity distortions are included [6, 10]. The strength of the R line is, therefore, directly related to the strength of the ${}^4\text{A}_2 \leftrightarrow {}^4\text{T}_2$ transition. That the R lines in absorption are much stronger for the γ centre than for α and β confirms that the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$, ${}^4\text{T}_1$ absorption bands are dominated by absorption from Cr^{3+} ions at centre γ .

The results of these experiments show that there are three major centres that contribute to the optical spectra of $\text{Cr}^{3+} : \text{LiNbO}_3$. The weak-field γ centre is dominant among these three. The decay of the broad-band emission excited in the ${}^4\text{T}_2$ absorption band and detected close to the peak of the ${}^4\text{T}_2$ emission band is dominated by the contribution from the γ centre which has a decay time of 10 μs . A small component of the decay has a lifetime of

about 300 μs , the decay time associated with the strong-field centres, α and β [11]. From the amplitudes of each component in the decay curve, the combined contribution of centres α and β to the broad-band intensity is estimated at less than 5%.

3.2. Site selective laser excitation spectroscopy

Site selective excitation and fluorescence line narrowing are useful tools with which to probe multisite systems and inhomogeneously broadened lines. Exciting the R_2 component of either centre α or β with a narrow-band laser gives rise to luminescence only from the R_1 component of the excited centre [9] showing that the Cr^{3+} ions at centres α and β do not form an exchange-coupled pair as was suggested by Weiyi *et al* [8]. A sequence of FLN spectra is shown in figure 4. When excited in the low-energy wing of the ${}^4A_2 \rightarrow {}^4T_2$ absorption (e.g. near $\lambda = 715$ nm) the same three-line pattern is observed as for excitation at higher energies in the same band. At these wavelengths no site selection occurs because the excitation band is vibronically broadened and all centres are equally excited. For excitation wavelengths between 723.5 nm and 725.5 nm there are four lines, three of which are independent of excitation wavelength. The fourth line, which shifts linearly with the laser energy, is due to the close proximity of the 4T_2 and 2E levels. At 723.5 nm all of the ions at centres α and γ are excited in the 4T_2 band as well as a fraction of the β centres, giving the same three lines as observed when exciting at 715 nm. In addition, a fraction of the β centres are excited in their R_2 line, thereby enhancing part of the corresponding R_1 line. The same effect is seen for excitation wavelengths around 728 nm: some ions at α centres are excited selectively via their R_2 line and non-selectively in their 4T_2 band. These experiments show clearly, once again, that the most prominent features of the optical spectra are due to three independent centres. The close proximity of the R lines to each other and to their respective 4T_2 bands show that the 4T_2 and 2E states are energetically close for all three centres. The crystal field is thus in the intermediate regime though it will be referred to as being strong or weak depending on the lower of the two states as described in the introduction.

There are also two additional, weak, non-resonant features that shift linearly with the laser excitation energy. These are offset from the laser excitation energy by 150 cm^{-1} and 240 cm^{-1} , corresponding to Raman spectra reported previously [30]. Whilst these features are not relevant to the present study it is important to know their origin in order to unravel the complicated spectra shown in figure 4.

3.3. Measurement of ground state splittings

The 4A_2 ground state is split by non-octahedral components of the crystal field. A trigonal distortion partially lifts the spin degeneracy to give a doublet ground term, ${}^4A_2(\bar{E})$ and ${}^4A_2(2\bar{A})$, split by $|2D|$, the so-called zero-field splitting. The D parameter can be measured using electron spin resonance (ESR) and Burns *et al* [5] were the first to do this in $\text{Cr}:\text{LiNbO}_3$, reporting a single value of $|2D| = 0.90$ cm^{-1} . Subsequently, Rexford *et al* [12] reported more sensitive ESR studies, identifying three separate centres, one characterized by $|2D| = 0.822$ cm^{-1} , which was identified with the spectrum reported by Burns *et al* [5] on the basis of identical g values. They also reported another spectrum with $|2D| = 0.42$ cm^{-1} and a feature too weak for analysis.

No studies of $\text{Cr}:\text{LiNbO}_3$ have linked specific ESR spectra to corresponding features in the optical spectra. Optically detected magnetic resonance (ODMR) is a technique which joins the two spectroscopies. Ground and excited state populations perturbed by the

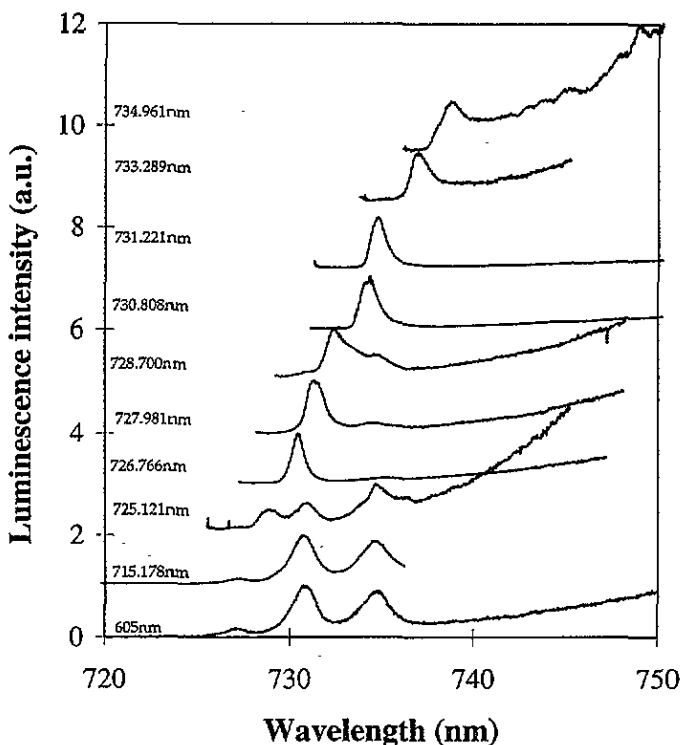


Figure 4. Non-resonant FLN spectra of $\text{Cr}^{3+}:\text{LiNbO}_3$ excited in the R line region by a Ti:sapphire laser. The selectively excited fluorescence widths are slit width limited.

application of a resonant microwave field will cause changes in the intensity of the optical spectrum. High-resolution FLN is analogous to ODMR in zero magnetic field. Under these circumstances, the ground state splitting observed can only correspond to the optical centre being excited, making it possible to relate the optical transition to the corresponding ESR spectrum.

The ground state splitting of centre α was measured by high-resolution FLN to be $|2D| = 0.84 \pm 0.02 \text{ cm}^{-1}$ (figure 5(a)). Within experimental error, this corresponds to the zero-field splitting $|2D| = 0.822 \text{ cm}^{-1}$, measured by ESR spectroscopy [5, 12]. The ground state splitting of the R(β) lines is found to be only slightly smaller than that of the R(α) lines, being given by $|2D| = 0.72 \pm 0.02 \text{ cm}^{-1}$. No such value has been reported in the ESR studies of various authors [5, 12–15]. The close similarity in the R line positions and splittings and of the measured value of $|2D|$ strongly suggest that centres β and α are related to the same crystallographic site perturbed in different ways by the disordered nature of the congruently grown $\text{Cr}^{3+}:\text{LiNbO}_3$ crystal. Note that the slightly greater width of the R(β) lines in figure 3(b) suggests that the optical inhomogeneous broadening is larger for centre β than for centre α . The associated ESR lines may then be expected to be similarly broadened, making them difficult to distinguish from the α centre lines with which they will almost overlap.

The measured ground state splittings at centres α and β are further evidence that these centres do not involve exchange-coupled pairs of Cr^{3+} as has been suggested [8, 13]. In the presence of exchange coupling the Hamiltonian for the two-ion system is classified

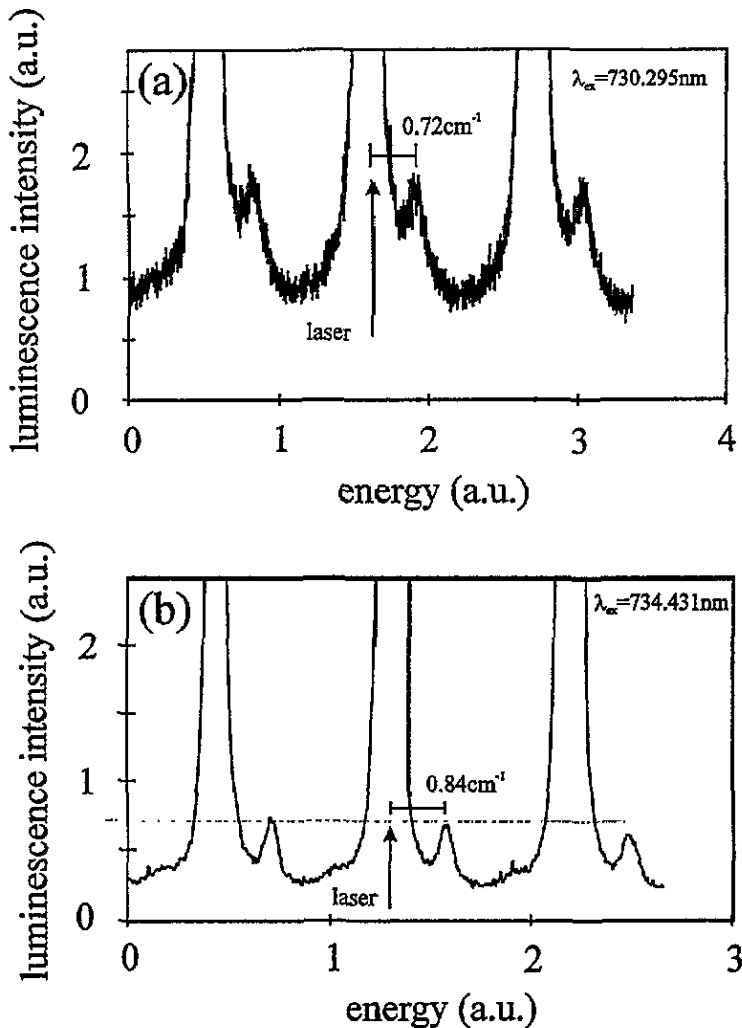


Figure 5. FLN spectra showing the ground state splitting of Cr^{3+} ions in (a) the β centre and (b) the α centre, measured at 15 K.

according to the total spin $S = S_1 + S_2$. For spin 3/2 ions S can be 0, 1, 2 or 3 and four transitions are commonly observed [17], split approximately by the Landé interval rule [10, 18]. Moreover, for Cr^{3+} concentrations greater than 0.2 at.%, the measured value of $|2D|$ for Cr–Cr pairs is 0.328 cm^{-1} [19], quite different from that of isolated ions. No evidence of R lines characterized by such values of total spin and ground state splitting has been found in the present study.

All previous ESR studies of $\text{Cr}:\text{LiNbO}_3$ have reported the same centre to be dominant, that with a ground state splitting of $|2D| \approx 0.8 \text{ cm}^{-1}$. It is therefore logical to expect that this spectrum corresponds to the dominant optical centre γ . The ground state splitting has not been measured optically since the $R(\gamma)$ lines only fluoresce at high temperatures where the homogeneous linewidth dwarfs the ground state splitting, rendering the FLN technique inoperable. However, Zhao *et al* [20] have calculated the ground state splitting using the method of Macfarlane [21, 22]. This calculation requires a knowledge of B , C , $10 Dq$, the

R line and ${}^4\text{T}_1$ splittings and the spin-orbit coupling parameter. The former four parameters have been discussed and quantified in subsection 3.1 and the ${}^4\text{T}_1$ splitting has been measured from polarized absorption spectra to be 700 cm^{-1} [7]. The spin-orbit coupling parameter is taken to be 210 cm^{-1} , a value that has been shown to be correct for other intermediate crystal field hosts [23, 24] and which is consistent with typical values reported for g_{\parallel} for all centres measured by ESR [25]. Zhao *et al* [20] obtained a value of $2D = 0.748\text{ cm}^{-1}$ but using the above parameters $2D$ is calculated to be -0.845 cm^{-1} , in good agreement with the magnitude of 0.822 cm^{-1} measured by ESR [12]. The same centre is therefore shown to dominate both the optical and ESR spectra.

The ${}^4\text{A}_2$ and ${}^2\text{E}$ splittings have been shown to be very similar for all three optically detected centres and so it is concluded that the α , β and γ centres all contribute to the dominant feature in the ESR spectrum. This lends further weight to the argument that the slightly perturbed β ESR spectrum is hidden in the wings of the much stronger combined contributions of the α and γ centres.

3.4. Spectroscopy of $\text{Cr}:\text{LiNbO}_3$ codoped with magnesium

Incorporation of Mg^{2+} ions into LiNbO_3 has significant influence on the optical properties of the material, including the spectroscopy of Cr^{3+} impurity ions. As figure 2 shows, the absorption spectrum of $\text{Cr}, \text{Mg}:\text{LiNbO}_3$ measured at 77 K has clear differences from that of $\text{Cr}:\text{LiNbO}_3$, mainly due to the broad absorption bands at 550 nm and 750 nm. These bands are broader than their counterparts in the $\text{Cr}^{3+}:\text{LiNbO}_3$ spectra; they are assigned to the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$ and ${}^4\text{A}_2 \rightarrow {}^4\text{T}_1$ transitions of a centre that will be referred to as δ . Their energies place the centre firmly in the weak-field regime.

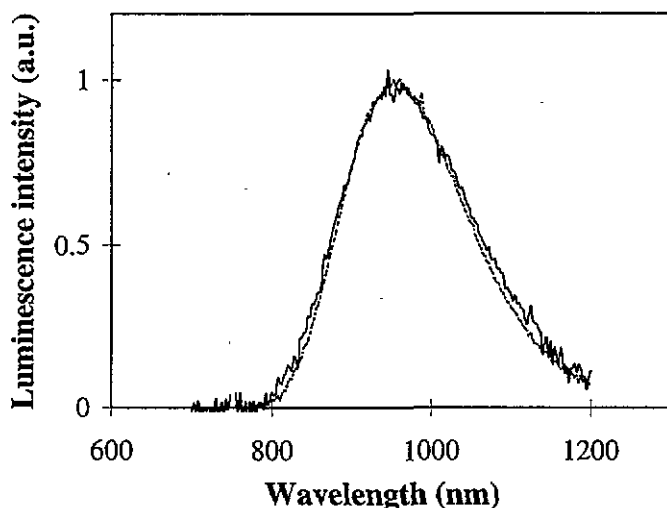


Figure 6. Normalized ${}^4\text{T}_2 \rightarrow {}^4\text{A}_2$ luminescence spectra in double-doped $\text{Cr}^{3+}, \text{Mg}^{2+}:\text{LiNbO}_3$ (dashed line), compared with that of $\text{Cr}^{3+}:\text{LiNbO}_3$. Excitation is at 760 nm in both cases.

The luminescence spectrum of $\text{Mg}, \text{Cr}:\text{LiNbO}_3$ is dominated by a broad ${}^4\text{T}_2 \rightarrow {}^4\text{A}_2$ emission band (figure 6). When excited at 760 nm in the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$ band, where the overlap with the absorption band of the γ centre is small, the luminescence spectrum is red shifted by 50 nm compared to the spectrum in figure 3, confirming the δ centre to have a weaker crystal field than the γ centre. The peak of the luminescence band does not depend

significantly on the excitation wavelength except where there are contributions from both centres, suggesting that there is minimal inhomogeneous broadening of the ${}^4A_2 \rightarrow {}^4T_2$ band of centre δ .

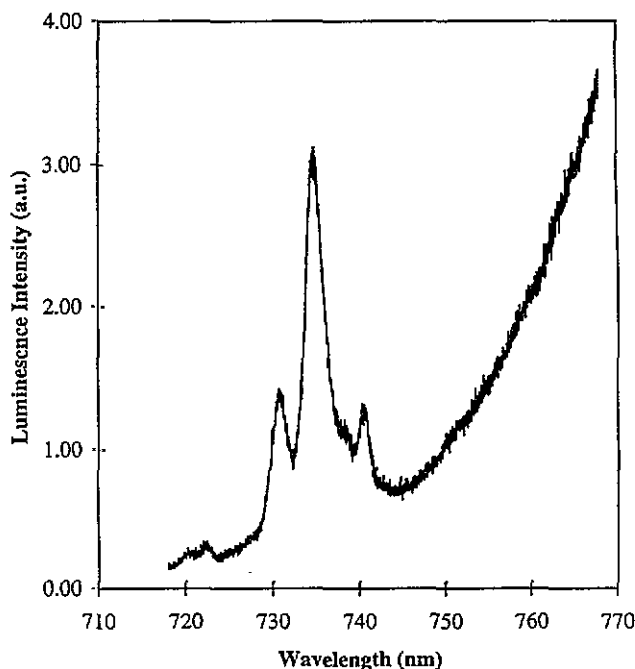


Figure 7. The R line luminescence in Cr^{3+} , Mg^{2+} : $LiNbO_3$, excited at 580 nm. The features at 737.5 nm and 740.5 nm are barely visible in the equivalent Cr^{3+} : $LiNbO_3$ spectrum.

There are also significant differences in the sharp-line luminescence of Mg , Cr : $LiNbO_3$ (figure 7). The lines at 740.5 nm and 737.5 nm have been identified at ${}^2E \rightarrow {}^4A_2$ transitions related to the δ centre [16]. If this were the case these R lines would not be observed at low temperature since the zero-phonon line of the ${}^4A_2 \rightarrow {}^4T_2$ absorption band with a peak at 750 nm should be near 800 nm, some 1000 cm^{-1} lower in energy than the 2E level. Thermal population of the 2E states should be small. In addition, the R lines of a weaker-field centre should be observed at higher photon energy than the R lines of the strong-field centres α and β . If these additional lines are R lines then they must be due to another centre, centre ϵ . In consequence, a total of five Cr^{3+} centres are required to interpret the optical spectrum of Mg , Cr : $LiNbO_3$.

Also shown in figure 6 is the luminescence spectrum of the Cr : $LiNbO_3$ sample excited at the same wavelength as the Mg , Cr : $LiNbO_3$ crystal. These spectra have the same lineshape though the intensity is weaker in the case of the Cr : $LiNbO_3$ sample. The R line at 739.1 nm has been observed in Cr : $LiNbO_3$ [26] and in the resonantly excited FLN spectra in figure 4 excited at wavelengths exceeding 735 nm. These results suggest that the effect of codoping $LiNbO_3$ with Mg^{2+} and Cr^{3+} is not to create additional centres associated with Cr - Mg complexes but rather to increase substantially the density of those centres already present in low concentrations in Cr : $LiNbO_3$.

3.5. Charge compensation and site occupation of Cr^{3+} in LiNbO_3

It has been shown above that there are contributions from five centres to the optical spectra of Cr^{3+} in LiNbO_3 . Centres α , β and γ all contribute to the dominant signal recorded in the ESR spectra. As the γ centre dominates the optical spectra it is reasonable to assume that the α and β centres are perturbed γ centres. Although the δ and ε centres increase in intensity with the addition of Mg^{2+} ions to the crystal they are also present in magnesium-free material. The physical nature of these five centres is now discussed.

Previous studies have taken both theoretical and experimental approaches. The nature of the dominant γ centre was first discussed by Glass [7]. Partially on the basis of a point ion calculation Glass proposed that Cr^{3+} substitutes preferentially at the Nb^{5+} site, and that the necessary charge compensation was by oxygen vacancies. Michel-Calendini *et al* [27] calculated the energies of Cr^{3+} in the Li^+ and Nb^{5+} sites using a molecular orbital model. They determined that when Cr^{3+} occupies the central position in the Li^+ octahedron then that site is slightly favoured over the equivalent position in the Nb^{5+} octahedron. However, Chang *et al* [28] stressed the importance of using the more precise C_3 site symmetry in their calculations rather than the simplifying assumption of C_{3v} symmetry used by others. They derived spin-Hamiltonian terms using superpositions of single ligand contributions and diagonalized the crystal field matrix to obtain the energy levels of Cr^{3+} in the Li^+ and Nb^{5+} sites. Their results appeared to support Cr^{3+} substitution at both sites. However, they claimed that only Cr^{3+} occupancy of the Nb^{5+} site accounted for the main ESR spectrum. In earlier studies the same claim had been made for the Li^+ site [29,30]. A cluster model analysis of the properties of the Cr^{3+} - Cr^{3+} pair complex proposed by Weiyi *et al* [8] supported the dimeric charge compensation mechanism [31], now discredited as discussed above and in [9].

Theoretical approaches that rely to a great extent on optimizing free parameters to fit experimentally determined energies fail to determine the site occupation of Cr^{3+} in the various sites in LiNbO_3 . This is unsurprising given that the R lines are rather insensitive to changes in the crystal field, that the Nb^{5+} and Li^+ ions have almost equal orbital radii, that the charge misfit is identical for Cr^{3+} entering either of the substitutional sites and that the sizes of the octahedral sites are similar. When the defect models and their energy level structure are based on incorrect interpretation of spectroscopic data [32] with good agreement between model and experiments still being claimed [28,29], the theoretical approach may be inappropriate. In reality the differences in the energy level structure of Cr^{3+} ions in the different locations in LiNbO_3 are so slight that the theoretical models cannot reliably distinguish between the particular spectral features of particular defect sites. In consequence, the best approach to the assignment of the site occupation of Cr^{3+} in LiNbO_3 is on the basis of the experimental results.

A purely experimental approach was taken by Rexford *et al* [12] who suggested that the three ESR spectra they observed corresponded to Cr^{3+} ions in the Nb^{5+} and Li^+ sites and the empty octahedron. Grachev *et al* [13] argued on the basis of ESR and ENDOR studies that the Cr^{3+} ions entered both the Li^+ and Nb^{5+} sites, charge compensation for Cr^{3+} at concentrations of less than 0.2% being by a pairing process involving Cr^{3+} ions and other neighbouring impurity cations (e.g. Al^{3+} or Fe^{3+}). The incorporation of Al^{3+} and Fe^{3+} ions into the crystal is accidental and different samples grown in different laboratories would not have the same concentrations of accidental impurities. All published optical spectra of $\text{Cr} : \text{LiNbO}_3$ are very similar so the different centres are unlikely to be due to this charge compensation mechanism. Nicholls *et al* [9] agreed that Cr^{3+} would occupy both the Li^+ and Nb^{5+} sites, arguing from the crystal-field calculations of Glass [7] that the R(α)lines

are due to $\text{Cr}^{3+}(\text{Li})$ sites and the $R(\beta)$ lines to $\text{Cr}^{3+}(\text{Nb})$ sites. All of these models have been proved to be wrong by the present study, which shows that the main Cr^{3+} centres in LiNbO_3 are associated with the same site.

Rutherford backscattering experiments in Nd-doped LiNbO_3 [33] have identified three inequivalent Nd^{3+} sites which are due to there being three stable positions for Nd^{3+} within the lithium octahedron, all different from the natural position of lithium. The Nd^{3+} ions then experience three different trigonal fields resulting in different splittings but the same barycentre for the transitions. The effect has been attributed to different methods of charge compensation. Such a mechanism does not seem likely for $\text{Cr}:\text{LiNbO}_3$. The α , β and γ centres have very similar ground and excited state splittings, suggesting that the trigonal field is approximately the same in each case. If the strong field R lines are associated with separate charge compensators their effect would rather be to change the Racah B parameter or apply an A_{1g} totally symmetric distortion to the octahedron.

The most comprehensive study of the ground state ESR spectra of $\text{Cr}^{3+}:\text{LiNbO}_3$ has been performed by Corradi *et al* [14]. The most intense ESR signal, corresponding to centre γ , was assigned from ESR and ENDOR studies to the $\text{Cr}^{3+}(\text{Li})$ site. The ability of the ENDOR technique to determine the nearest-neighbour ions of the impurity ion under investigation makes this study very reliable.

A natural mechanism for α , β and γ centres to be associated with the Li^+ site in LiNbO_3 is provided by the compositional disorder of the crystal, at least in crystals grown from the congruently melting composition. Nb^{5+} ions occupy 5.9% of Li^+ sites and a slightly smaller fraction (4.7%) of Nb^{5+} vacancies are required for charge compensation. These intrinsic imperfections in LiNbO_3 occur in far greater concentrations than the most common impurities [34]. Changes in the charge distribution caused by the defects influence the energy levels of the Cr^{3+} ion. It is reasonable to expect that defects in the second-nearest-neighbour shell (a cation shell) will have the strongest effect. The Li^+ site, shown in figure 1, is surrounded by an approximate octahedron of Nb^{5+} sites and a slightly larger, rotated octahedron of Li^+ sites. With the Nb^{5+} site along the c axis at approximately the same distance this gives seven second-nearest-neighbour Nb^{5+} ions and six Li^+ ions. In consequence, it is expected that some Cr^{3+} ions have a neighbouring Nb^{5+} vacancy or an Nb^{5+} antisite as occasional alternatives to the perfect lattice configuration. Assuming a Poissonian distribution of defects, 49% of Cr^{3+} ions in Li^+ sites should occupy sites with no second-nearest-neighbour defects, 17% should have one Nb vacancy and 18% one Nb antisite. The almost equal intensities of $R_1(\alpha)$ and $R_1(\beta)$ lines suggest that the numbers of Cr^{3+} ions in the α and β centres are approximately equal. The actual intensities of the $R_1(\alpha)$ and $R_1(\beta)$ lines are weaker than would be expected from a purely statistical occupation hypothesis but the simple model does predict three dominant centres, one of which is much more common than the other two almost equally populous perturbed sites. The proportion of α and β centres is higher than would appear from a simple inspection of the fluorescence spectra because these centres also decay via the broad band (see subsection 3.1), thus reducing the intensities of their R lines. Furthermore, there is evidence that defects are not entirely free and tend to form clusters, thus reducing the number of isolated defects [34]. Hence, it would be expected that the proportion of Cr^{3+} ions occupying defect centres would be reduced. It is concluded therefore that the γ centre is the undistorted Li^+ site and the α and β centres are γ centres distorted by the presence of either a Nb^{5+} vacancy or antisite.

The addition of Mg^{2+} causes increased occupation of the δ and ε optical centres. In a comparative study of $\text{Cr}:\text{LiNbO}_3$, Cr , $\text{Mg}:\text{LiNbO}_3$ and ruby, Thiemann *et al* [15] reported an additional Cr^{3+} centre with isotropic ESR spectrum ($D < 0.01 \text{ cm}^{-1}$) to be present in

LiNbO₃ co-doped with Cr³⁺ and Mg²⁺ and that this was responsible for the additional absorption bands. ENDOR results had already proved that Cr³⁺ could substitute in a Nb⁵⁺ site [14]. The authors believed that the ENDOR parameters (hyperfine splitting, quadrupole splitting, axial field splitting) were so different from those of the centre with $|2D| = 0.822 \text{ cm}^{-1}$ that the two centres relate to different sites. Thiemann *et al* [15] inferred that in Cr:LiNbO₃ the Cr³⁺ ion must substitute for Li⁺ and that the addition of Mg²⁺ forces Cr³⁺ into the Nb⁵⁺ site. This suggestion does not conflict with the proposals made here. By analogy with the Li⁺ sites, it is thought that the weak-field δ centre is the undistorted Nb⁵⁺ site while the strong-field ϵ centre is a Nb⁵⁺ site perturbed by a nearby vacancy or antisite. A summary of parameters relating to all five centres is given in table 2.

Table 2. A summary of the site occupation and energy levels of the Cr³⁺ centres in LiNbO₃.

Centre	² E energy (cm ⁻¹)	² E splitting (cm ⁻¹)	Broad-band fluorescence peak (cm ⁻¹)	Site occupation
α	13 640	95	n/a	Li (distorted)
β	13 720	64	n/a	Li (distorted)
γ	13 790	38	11 100	Li
δ	not observed	not observed	10 500	Nb
ϵ	13 532	55	n/a	Nb (distorted)

4. Conclusions

The history of Cr:LiNbO₃ spectroscopy has been shrouded in confusion. Until this study no authors had recognized the fact that three centres are required to account for the R lines that they had observed [5, 7–9, 16, 26, 28, 29, 32]. There were suggestions that R lines are hot-phonon side bands and exchange-coupled pairs [8]. In other reports R lines have been paired with partners having wildly different absorption and fluorescence characteristics [32]. The further study of Mg, Cr:LiNbO₃ has added to the confusion. It has been supposed that the Mg²⁺ ions create a new optical centre and that this centre emits through R line and broad-band fluorescence despite the ²E state being about 1000 cm⁻¹ above the ⁴T₂ state [16].

The present study has resolved the difficulties in interpretation of the spectra. The results of many optical and ESR studies have been integrated through the optical measurement of the ground state splittings of the main centres in Cr:LiNbO₃. It has been shown that the principal centres in Cr:LiNbO₃ are all the result of Cr³⁺ substitution at Li⁺ sites. The main centre that dominates both ESR and optical spectra is the undistorted Li⁺ site but R line fluorescence derives from Li⁺ sites that are perturbed by nearby Nb vacancies or antisites. The addition of Mg²⁺ to the lattice causes Nb⁵⁺ sites to be more preferentially occupied and a weak and a strong field centre relating to Cr³⁺ occupation of the Nb⁵⁺ site become more predominant in the optical spectra. The Nb⁵⁺ sites are weaker field sites than the Li⁺ sites and so it must be assumed that long-range charge compensation is more easily achieved for the Li⁺ sites, causing them to be preferred. The addition of Mg²⁺ to the lattice alters the fine balance in favour of the Nb⁵⁺ sites. It is likely that there are other much weaker features in the optical and ESR spectra of Cr:LiNbO₃ that relate to Li⁺ and Nb⁵⁺ sites perturbed by other combinations of antisites and vacancies. However, the five main optical centres have now been accounted for.

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